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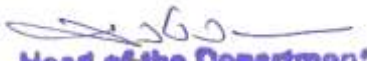
YELLAMANCHILI - 531055, ANDHRA PRADESH



DEPARTMENT OF PHYSICS

Publications, Seminars and Conferences Attended

S.No	Date	Name of the Lecturer	Event in which participated	Institute which organized the event
1.	27-02-2023	Sri K Venkanna	National seminar	Andhra Loyala college
2.	22-02-2023	Sri K Venkanna	International seminar	Andhra Loyala college
3.	25-03-2023	Sri K Venkanna	National Conference	Vignan university
4.	03-03-2023	Sri K Venkanna	National Seminar	GDC(W), Srikakulam
5.	01-02-2020	Sri K Venkanna	National Conference	GDC(W), Srikakulam
6.	14-12-2022	Sri K Venkanna	National Science academia workshop	M R College, Vizianagaram
7.	19-12-2014	Sri P S Jaggarao	National Seminar	Dr V S K GDC
8.	05-02-2015	Sri P S Jaggarao	National Level Workshop	Department of nuclear physics, AU
9.	06-07-2015	Sri P S Jaggarao	UGC Workshop	Dr V S K GDC
10.	10-10-2015	Sri P S Jaggarao	AP GCGTA Workshop	Andhra university
11.	04-08-1992	Sri P S Jaggarao	Dielectric and Resistive properties of lanthanum doped Ba Cu _{1/3} Ta _{2/3} O ₃	Indian journal of pure and applied physics
12.	10-04-1994	Sri P S Jaggarao	Dielectric and conductivity properties of lanthanum modified strontium copper niobate	Journal of material science letters
13.	30-06-2021	Sri B Nageswara Rao	https://doi.org/10.1016/j.matpr.2021.06.301	Materials Today
14.	12-07-2021	Sri B Nageswara Rao	https://doi.org/10.1016/j.matpr.2021.07.122	Materials Today


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Sri K Venkanna , Lecturer in Physics Presented Paper in “National seminar” held at Andhra Loyala college, organized by department of Physics Vijayawada dated: 27-02-2023.



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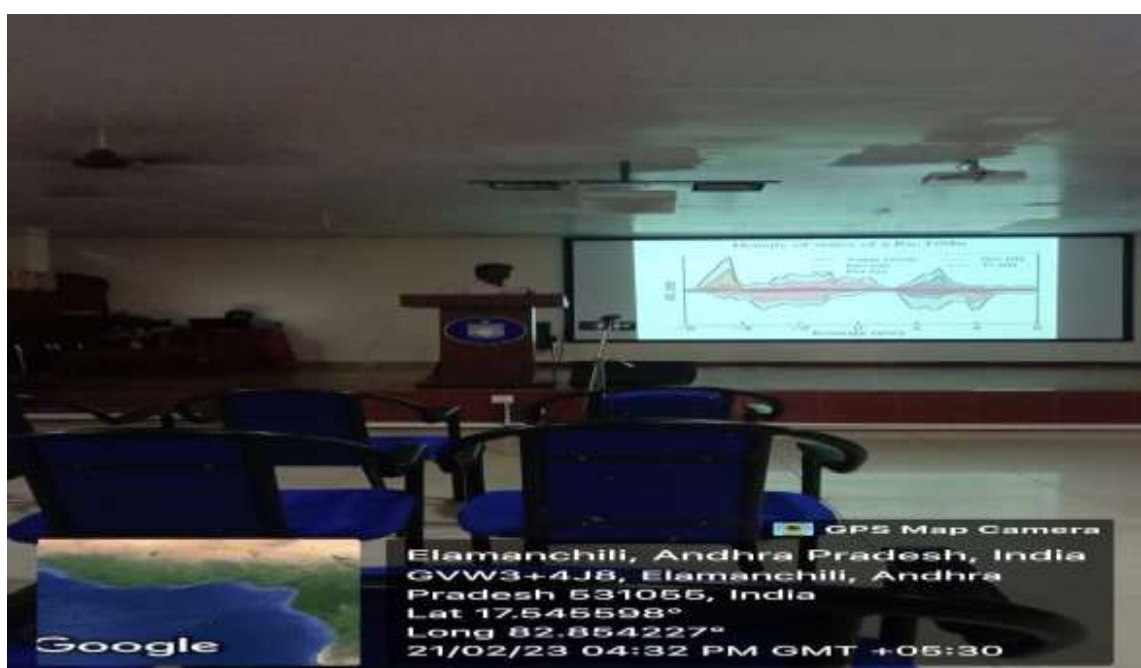
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Sri K Venkanna , Lecturer in Physics Presented Paper in “International seminar” held at Andhra Loyala college, organized by Department of Electronics, Vijayawada dated: 22-02-2023.



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Proceedings of ICESSET - 2023

First Principle Studies on Novel Ru₂TiMn Heusler Alloy for Thermoelectric Properties

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Abstract

We report the structural stability, electronic, magnetic, thermodynamic and thermoelectric properties of novel Ru₂TiMn Heusler alloys using first principle studies based on density functional theory (DFT). With the reports of martensitic phase transformation from cubic to tetragonal structural phase in this series of samples, Ru₂TiMn alloy is found to be stable in tetragonal phase at c/a ratio 0.92, fitted with Birch-Murnaghan equation of state. The observed negative formation energy making it the alloy is physically synthesizable. Further non-zero density of states (DOS) in both spin-up and spin-down channels near to Fermi energy with no energy gap proves its metallic character. Absence of imaginary frequencies in phonon dispersion spectrum confirms the dynamical stability of the compound. Integration of DOS indicates the magnetic moment of a compound and estimated magnetic moment is $3.025\mu_B$. It is observed that major contribution for magnetic moment is coming from Mn atoms. The calculations of different thermodynamic properties such as entropy, free energy and specific heat at constant volume (C_V) of Ru₂TiMn alloy are performed by varying the temperature from 0 K to 2000 K. To estimate thermoelectric properties of Ru₂TiMn alloy as a function of chemical potential, we have used the BoltzTrap code based on the Boltzmann model via Boltzmann transport equation. The figure of merit (zT) is very large as calculated thermal conductivity is low and Seebeck coefficient is large value of about $100 \mu V/K$, shows that our Ru₂TiMn alloy can be used for powerful thermoelectric devices.

Key words: Heusler alloy, martensitic phase, thermoelectric materials, dynamical stability



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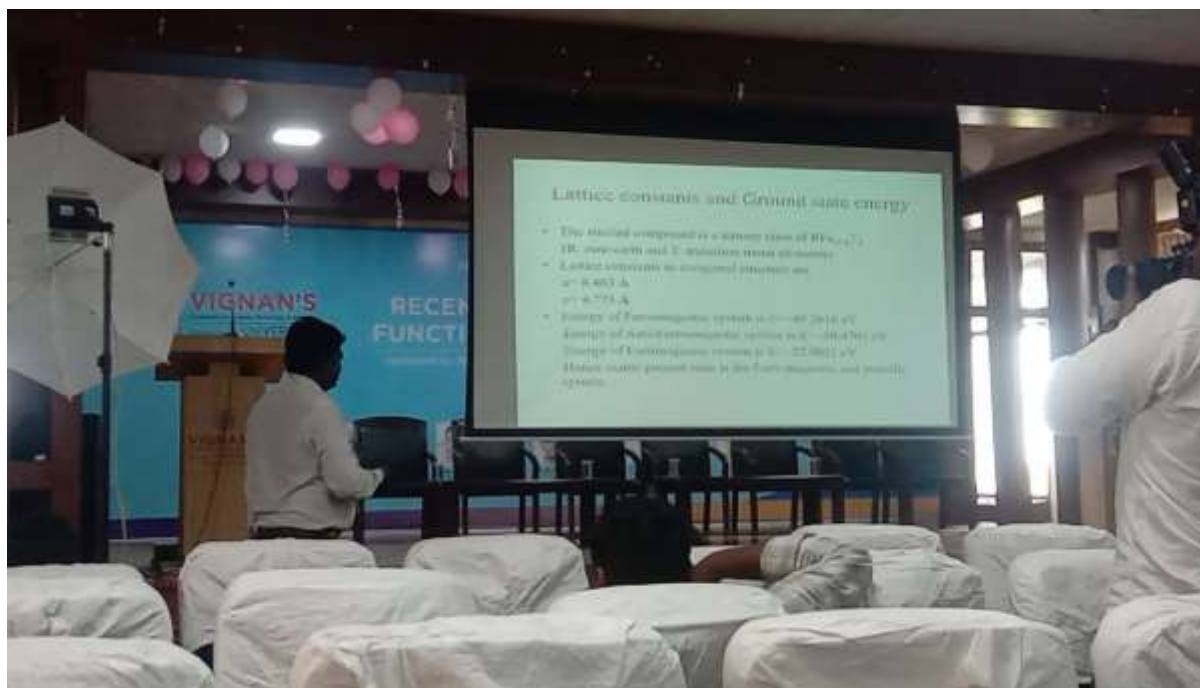
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Sri K Venkanna , Lecturer in Physics Presented Paper in “National Conference” held at Vignan university, organized by Department of Physics (BS), Guntur dated: 25-03-2023.



First principle studies on Iron rich permanent magnetic compound $\text{YFe}_{10}\text{Cr}_2$

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Abstract

We report on electrical, magnetic, thermoelectric and thermodynamic properties of iron rich, permanent magnetic compound $\text{YFe}_{10}\text{Cr}_2$ using first principles calculations. The studied compound is a ternary class of $\text{RFe}_{12-x}\text{T}_x$ (R: rare-earth and T: transition metal elements) family. For all the theoretical calculations, we have taken practically reported lattice constants in tetragonal structure as $a = 8.463 \text{ \AA}$ and $c = 4.775 \text{ \AA}$. From the density functional theory (DFT) studies, it is proved that the ground state of the compound is ferrimagnetic with metallic states through the structural optimization and spin density of states plots. The calculated magnetic moment of the compound is found to $\sim 16.7 \mu_B$ is matches well to the experimental value of $16.7 \mu_B^{23}$. It is observed from the spin density of states that the major contribution for the magnetic moment is from Fe-3d orbital states. The absence of negative phonons in a phonon dispersion spectrum suggest that the studied compound is dynamically stable. We have calculated exchange interaction constants (J_{ij}) using Green's function formalism as implemented in SPRKKR package. Using these interaction constants, it is estimated magnetic transition temperature as $T_C = 623 \text{ K}$, using mean field approximation which is close to the experimental value $T_C = 510 \text{ K}$. Further, to characterize the thermoelectric property, it is estimated the figure of merits are 0.4 % and 1.1 % at 1000 K and 2000 K, respectively.

Keywords

Iron rich permanent magnetic, Mean field approximation, Exchange interactions, Magnetic and thermoelectric properties.



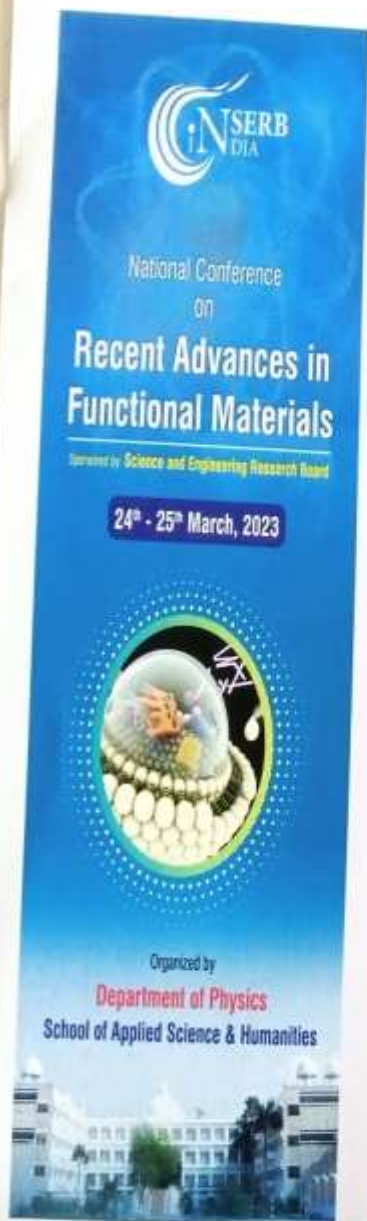
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Certificate

This is to certify that Prof./Dr./Mr./Ms. K. Venkanna, NIT, AP
has participated / presented a oral / poster presentation on First principle studies
on Iron rich permanent magnetic compound $YFe_{10}Cl_2$
in National Conference on "Recent Advances in Functional Materials" Sponsored by Science
and Engineering Research Board (SERB), Govt, New Delhi. Organized by Department of Physics,
School of Applied Sciences, VFSTR (Deemed to be University) during 24th - 25th March 2023.

Dr. V. Malapati
Convener

Prof. M. Sreenivasulu
Head, Physics

Prof. N. Srinivasu
Dean, AS&H

Prof. P. Nagabhusham
Vice-Chancellor



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Sri K Venkanna , Lecturer in Physics Attended a “National Seminar” held at GDC(W), Srikakulam dated: 03-03-2023.



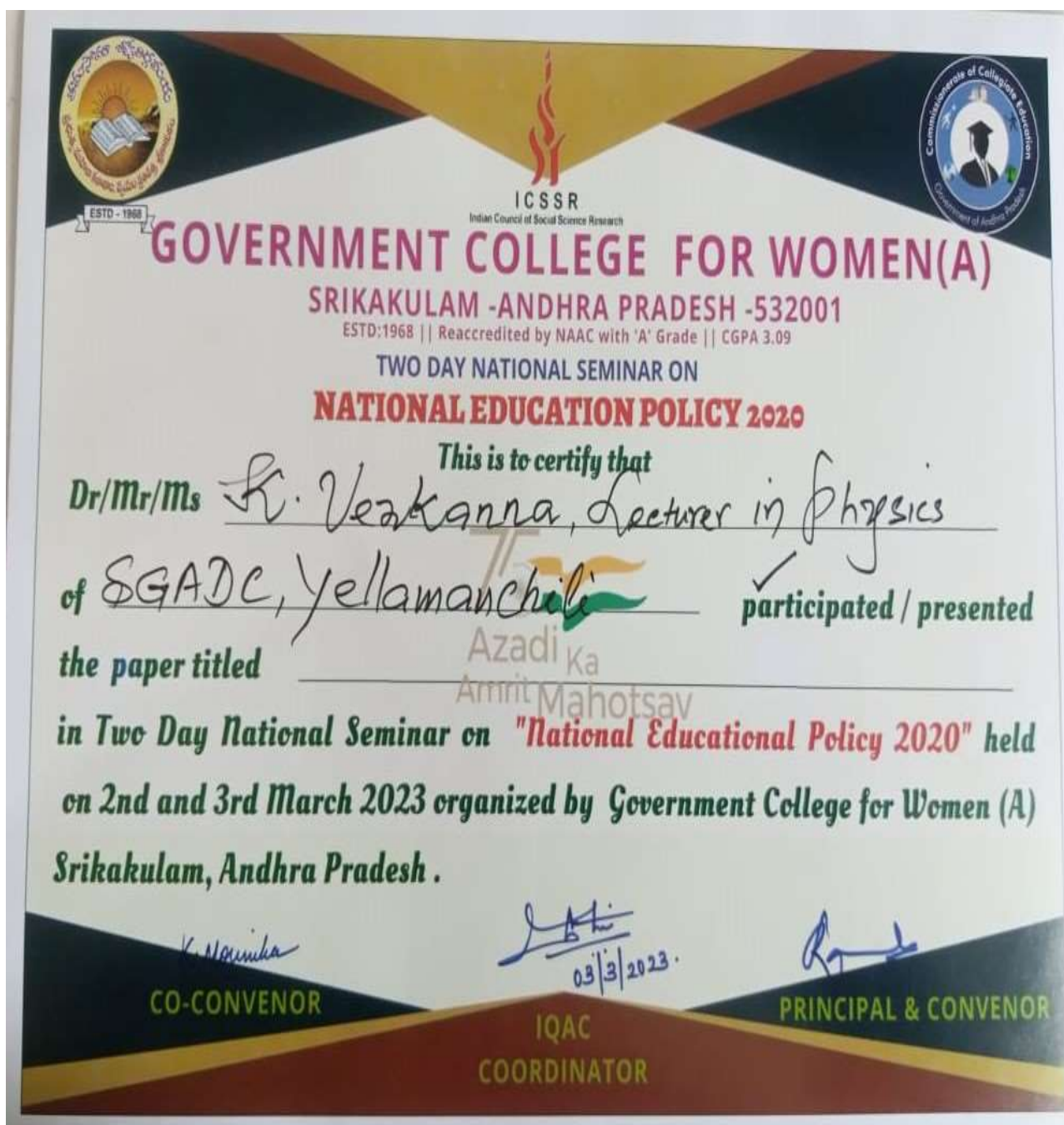
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Sri K Venkanna , Lecturer in Physics Attended a "National Conference" held at GDC(W), Srikakulam dated: 01-02-2020.



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Sri K Venkanna , Lecturer in Physics Attended a “National Science academia workshop” held at M R College, Vizianagaram dated: 14-12-2022.



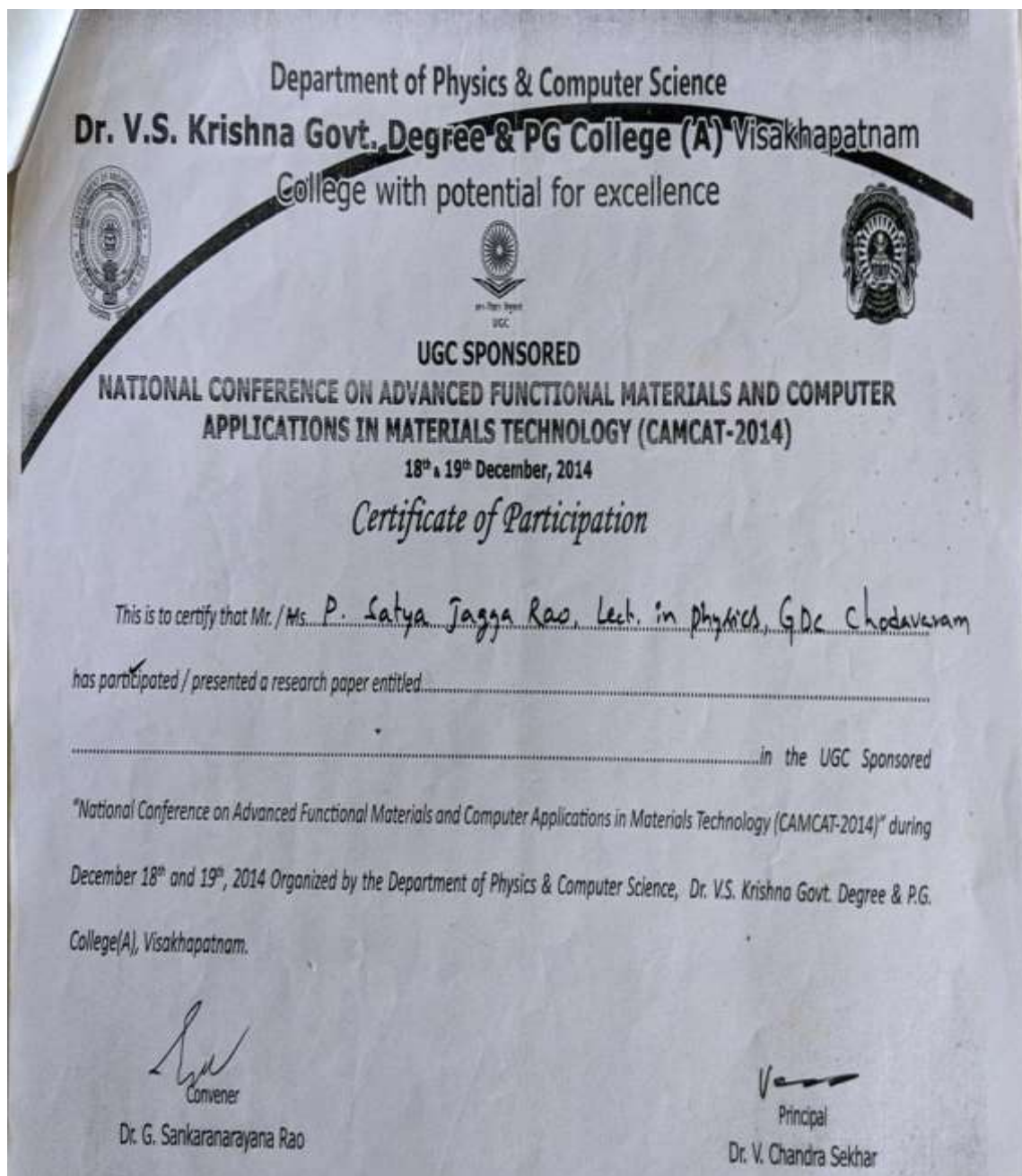
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Sri P S Jaggarao, Lecturer in Physics participated in "National Seminar" held at Dr V S K GDC, Visakhapatnam dated: 19-12-2014



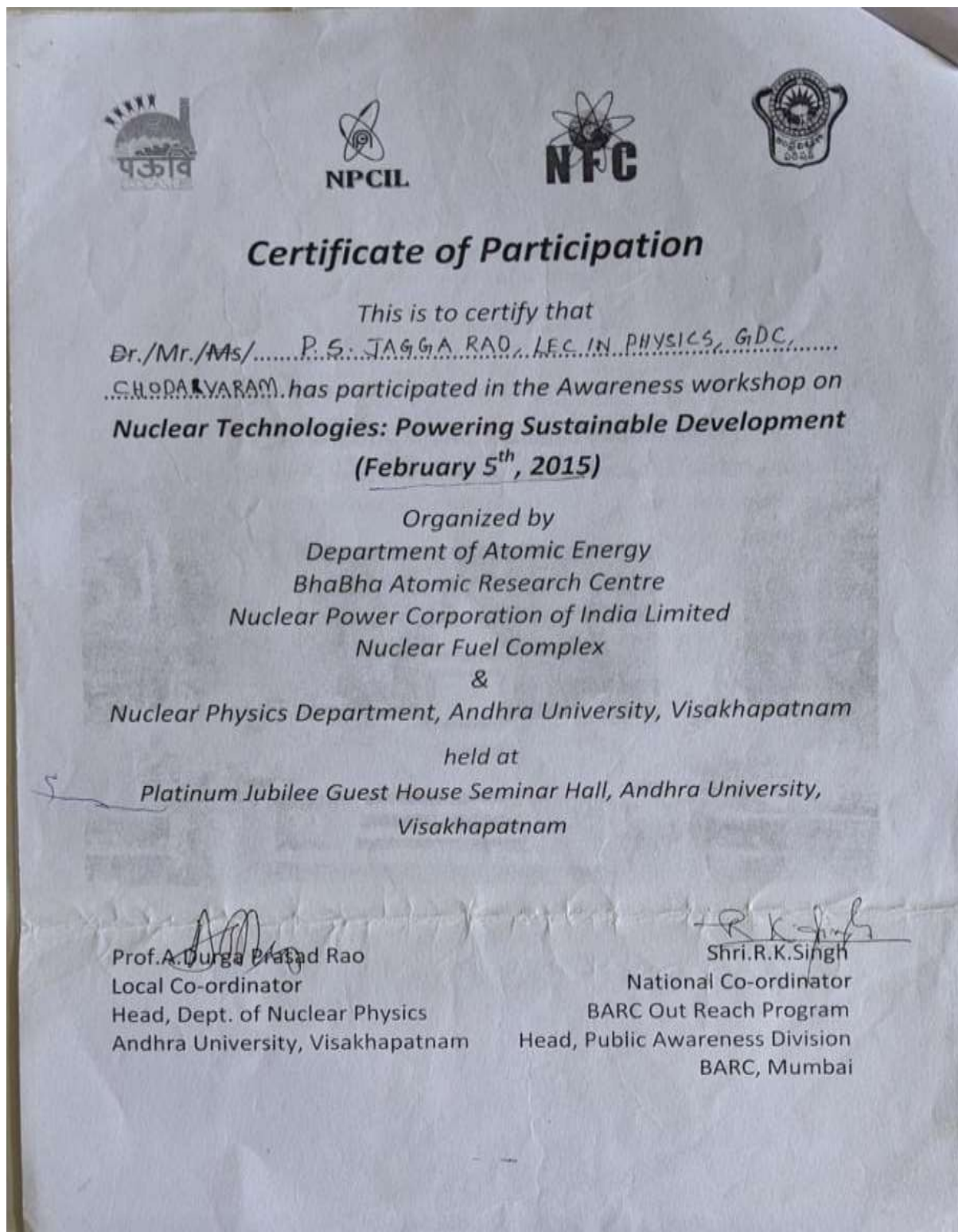
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Sri P S Jaggarao, Lecturer in Physics participated in “National Level Workshop” held at Andhra university, Organized by Department of Nuclear Physics, Visakhapatnam dated: 05-02-2015.



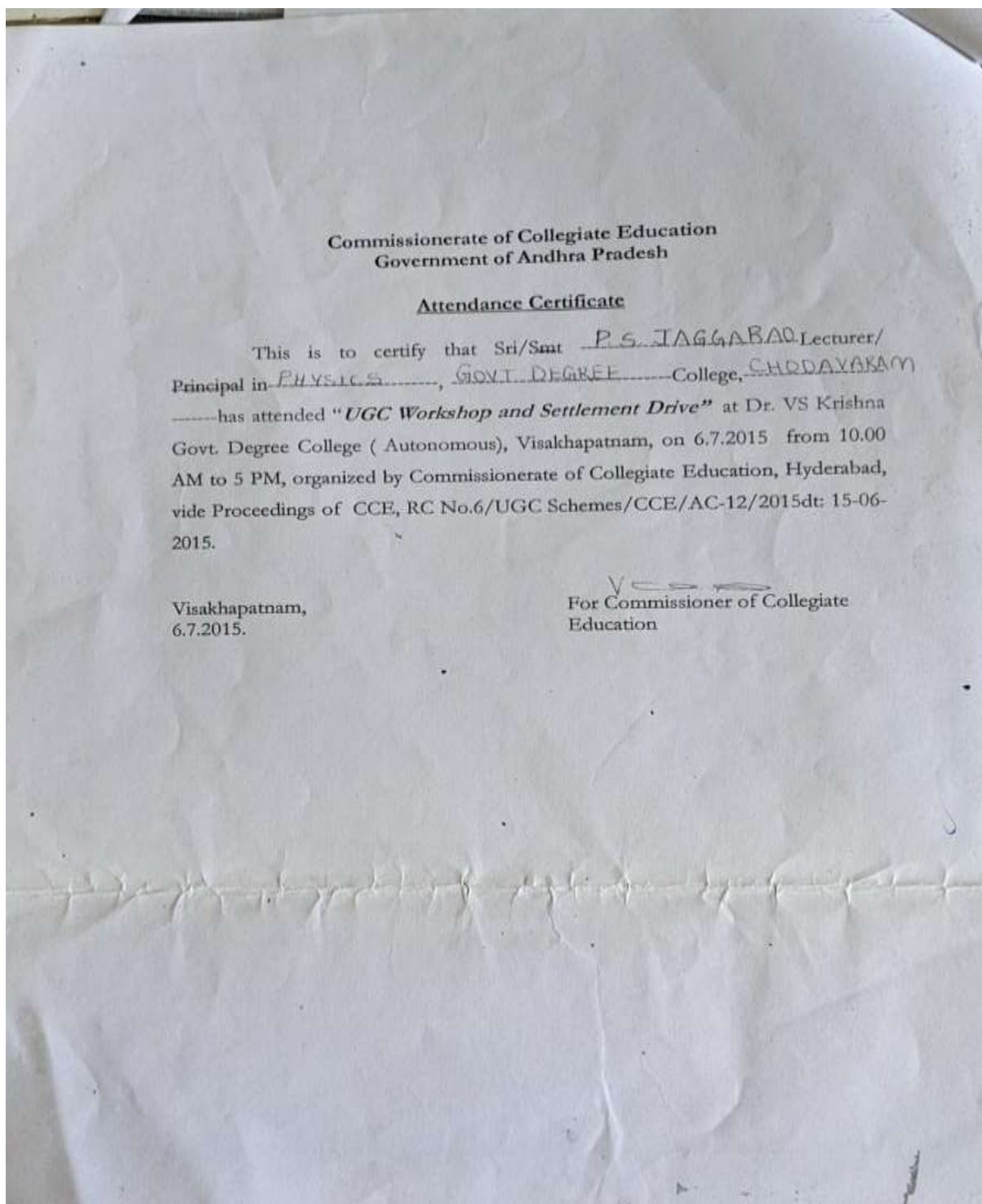
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Sri P S Jaggarao, Lecturer in Physics participated in "UGC Workshop" held at Dr V S K Government degree college, Visakhapatnam dated: 06-07-2015.



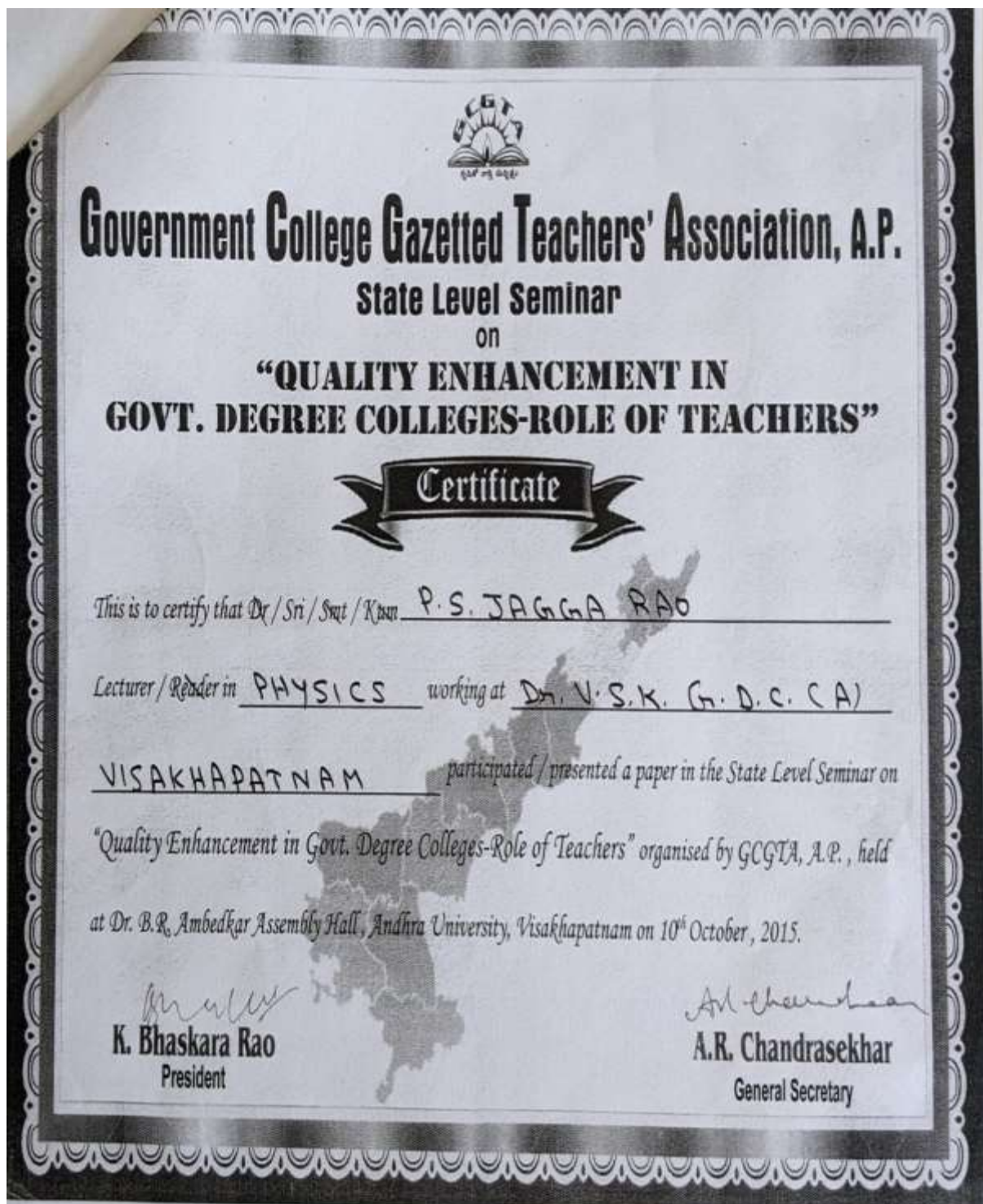
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Sri P S Jaggarao, Lecturer in Physics participated in “AP GCGTA Workshop” held at Andhra university, Visakhapatnam dated: 10-10-2015.



Full length papers presented by Sri P S Jaggarao-Paper-1

Indian Journal of Pure & Applied Physics
Vol. 31, January 1993, pp. 43-47

**Dielectric and resistivity properties of lanthanum doped
 $\text{Ba}(\text{Cu}_{1/3}\text{Ta}_{2/3})\text{O}_3$ and $\text{Ba}(\text{Cu}_{1/3}\text{Nb}_{2/3})\text{O}_3$**

K Sambasiva Rao, P S Jagga Rao & K Rama Rao

Department of Physics, Andhra University, Visakhapatnam 530 003

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School of Chemistry, Andhra University, Visakhapatnam 530 003

Received 27 April 1992; revised received 4 August 1992

A series of samples of the systems $\text{Ba}_{1-x}\text{La}_x(\text{Cu}_{1/3}\text{Ta}_{2/3})\text{O}_3$ and $\text{Ba}_{1-x}\text{La}_x(\text{Cu}_{1/3}\text{Nb}_{2/3})\text{O}_3$ where $x = 0.000, 0.025, 0.050, 0.075$ and 0.100 have been prepared by the usual ceramic technique. X-ray diffraction data indicated that they are of tetragonal symmetry. Dielectric properties of $\text{Ba}(\text{Cu}_{1/3}\text{Ta}_{2/3})\text{O}_3$ (BCT) revealed a diffused phase transition. Curie temperatures obtained from dielectric measurements were compared with differential thermal analysis studies. Doping of lanthanum in BCT and $\text{Ba}(\text{Cu}_{1/3}\text{Nb}_{2/3})\text{O}_3$ (BCN) decreased the curie temperatures of BCT and BCN. DC resistivity studies in modified and unmodified BCT and BCN indicated a positive temperature coefficient (PTC) behaviour.

1 Introduction

The discovery of barium titanate as ferroelectric material stimulated a further search and creation of perovskite-type and oxygen-octahedra ferroelectrics. Time has shown barium titanate appears as "a parent" of a huge family of ferro and antiferroelectrics with perovskite-type crystal structure. Perovskite-type ferroelectrics are found to be outstanding materials in view of their device applications, e.g. BaTiO_3 , lead zirconate titanate (PZT). Some copper containing perovskites $\text{Ba}(\text{Cu}_{1/3}\text{Ta}_{2/3})\text{O}_3$ and $\text{Ba}(\text{Cu}_{1/3}\text{Nb}_{2/3})\text{O}_3$ have been synthesised by Kapyshev *et al.*¹, Blasse² and Venevtsev³, respectively. Also, they were reported to be ferroelectric having high curie temperatures of 470 and 380°C respectively, with tetragonal symmetry. Literature survey revealed that no further experimental investigations have been made on these materials. Systematic dielectric and resistivity studies have been made on these materials for the first time by the present authors.

Present paper describes preparation, dielectric and resistivity studies on lanthanum modified barium copper tantalate (BCT) and barium copper niobate (BCN) systems.

2 Material Preparation

The constituent raw materials, BaCO_3 , CuO , Ta_2O_5 , Nb_2O_5 and La_2O_3 of reagent grade have

been weighed in required proportion to give the following compositions:

1.	$\text{Ba}(\text{Cu}_{1/3}\text{Ta}_{2/3})\text{O}_3$	BCT
2.	$\text{Ba}_{0.925}\text{La}_{0.075}(\text{Cu}_{1/3}\text{Ta}_{2/3})\text{O}_3$	0.05 La-BCT
3.	$\text{Ba}_{0.850}\text{La}_{0.150}(\text{Cu}_{1/3}\text{Ta}_{2/3})\text{O}_3$	0.10 La-BCT
4.	$\text{Ba}_{0.775}\text{La}_{0.225}(\text{Cu}_{1/3}\text{Ta}_{2/3})\text{O}_3$	0.15 La-BCT
5.	$\text{Ba}_{0.700}\text{La}_{0.300}(\text{Cu}_{1/3}\text{Ta}_{2/3})\text{O}_3$	0.20 La-BCT
6.	$\text{Ba}(\text{Cu}_{1/3}\text{Nb}_{2/3})\text{O}_3$	BCN
7.	$\text{Ba}_{0.925}\text{La}_{0.075}(\text{Cu}_{1/3}\text{Nb}_{2/3})\text{O}_3$	0.05 La-BCN
8.	$\text{Ba}_{0.850}\text{La}_{0.150}(\text{Cu}_{1/3}\text{Nb}_{2/3})\text{O}_3$	0.10 La-BCN
9.	$\text{Ba}_{0.775}\text{La}_{0.225}(\text{Cu}_{1/3}\text{Nb}_{2/3})\text{O}_3$	0.15 La-BCN
10.	$\text{Ba}_{0.700}\text{La}_{0.300}(\text{Cu}_{1/3}\text{Nb}_{2/3})\text{O}_3$	0.20 La-BCN

Ceramic samples have been prepared by the usual sintering technique. Actual procedure adopted for the preparation of materials is depicted in Fig. 1.

3 Experimental Techniques

The dielectric measurements were carried out at 1 kHz using DIGITAL LCR METER TYPE VLCR 6. The temperature is monitored by a Cr-Al thermocouple. The dc resistivity was measured by two-probe method using Keithley Model 614 Electrometer.

Powder X-ray diffraction (XRD) patterns of these materials have been obtained with the diffractometer at RSIC, Nagpur University, Nagpur.

*National Physical Laboratory, New Delhi 110 012



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Weighting (BaCO_3 , CuO , Ta_2O_5 ,
 Nb_2O_5 , La_2O_3)

Mixing (in alcohol, 4 hr)

Drying

Calcination ($950^\circ\text{C}/4$ hr twice)

Mixing with binder (5% PVA)

Pressing pellets (dia. = 15 mm, $t = 1.1$ mm
for BCT)
(dia. = 14 mm, $t = 1.7$ mm
for BCN)

Binder burn out ($450^\circ\text{C}/1$ hr)

Firing ($1200^\circ\text{C}/4$ hr)

Polishing (3 μm Al_2O_3 powder)

Electroding (air dry silver paste
cured at $1400^\circ\text{C}/1$ hr)

Fig. 1 - Procedure for sample preparation

4 Results and Discussion

4.1 Dielectric properties

The X-ray diffraction studies in the above materials, labelled 1-10, showed single-phase final product with tetragonal symmetry throughout the unmodified and modified compositions. The lattice parameters a and c as computed from the XRD data are shown in Table 1, which are in good agreement with the values reported in literature for BCT and BCN³.

Dielectric constant as a function of temperature has been measured at 1 kHz in the temperature range between room temperature and 450°C . For dielectric and resistivity measurements silver coated disc samples having dimensions around 1.1 mm thickness and 15 mm diameter of BCT, 14 mm diameter and 1.7 mm thickness of BCN cured at 400°C for one hour are used.

Fig. 2 illustrates the variation of dielectric constant with temperature for BCT and lanthanum modified BCT. A maximum dielectric constant of 179 was obtained for BCT at 319°C indicating its curie temperature. The observed curie temperature is much below the value of 470°C reported by Venetsev¹.

Differential thermal analysis studies have been made on BCT and the curie temperature is obtained from the peak temperature of exotherm (T_D). The sample BCT exhibited 300.1°C as the curie temperature. However, the curie tempera-

Table 1 - Lattice parameters of La-modified BCT and BCN

S.No.	Composition	$a(\text{\AA})$	$c(\text{\AA})$
1.	BCN	4.0804	4.1397
2.	0.05 La-BCN	4.0424	4.1853
3.	0.10 La-BCN	4.0611	4.1639
4.	0.15 La-BCN	4.0756	4.1695
5.	0.20 La-BCN	4.0533	4.1307
6.	BCT	4.025	4.2359
7.	0.05 La-BCT	4.0156	4.2147
8.	0.10 La-BCT	4.0416	4.1618
9.	0.15 La-BCT	4.0260	4.1980
10.	0.20 La-BCT	4.0350	4.1800

ture obtained from the dielectric measurement is 319°C . A typical DTA trace has been shown in Fig. 3. Similarly, in the case of 0.05 La-BCT the curie temperature obtained from the dielectric measurement is 270°C , while the value obtained from DTA is 297.3°C .

As the concentration of lanthanum in BCT increases, the curie temperature decreases from 319 to 199°C and the peak dielectric constant increases with lanthanum content except for 0.20 La-BCT (Fig. 2). Similar behaviour has been ob-

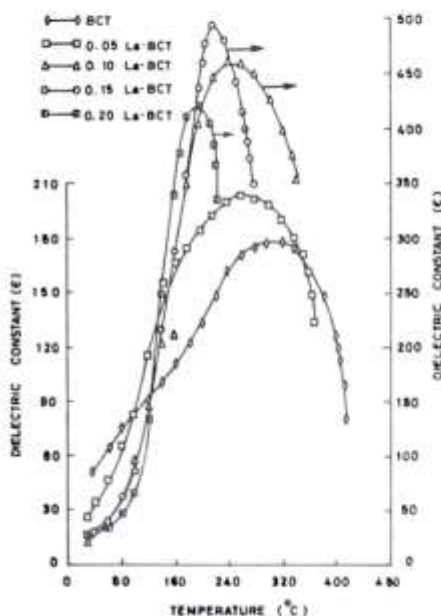


Fig. 2 - The variation of dielectric constant with temperature for modified and unmodified BCT

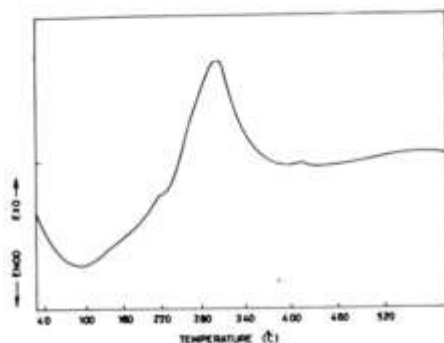


Fig. 3 - DTA trace of BCT

served in different perovskite systems, for example lead zinc niobate-lead titanate-barium titanate⁴ system, barium titanate zirconate⁵ and lanthanum doped lead titanate systems⁶.

Further, it is evident that a diffused phase transition (DPT) is observed in the compositions from 1 to 5. In para region i.e., above T_c and equation⁷ of the following form has been obeyed:

$$\frac{1}{\epsilon} = \frac{1}{\epsilon_T} + \frac{(T - T_c)^2}{2\epsilon_T\delta^2}$$

where ϵ_T and δ are maximum dielectric constant and intensity of diffusivity, respectively. The δ values for compositions 1-5 were computed to be 70, 73, 73.5, 52 and 40 K respectively, which indicate a decrease in diffusivity with increase in lanthanum concentration in BCT.

The dielectric constant as a function of temperature in BCN and lanthanum doped BCN has been shown in Fig. 4. It is clear that at 400°C a maximum dielectric constant is observed in unmodified barium copper niobate. This transition temperature is very much nearer to the value of 380°C reported on BCN by Venetsev³. The transition temperatures noted from the DTA traces in case of BCN and 0.05 La-BCN are 426.3 and 289°C, respectively. Though the transition temperatures T_c and T_D for both the systems BCT and BCN reflect the same trend, they do not coincide exactly. The reason for this discrepancy is under investigation.

It can be noticed from Fig. 4 that lanthanum modification in BCN facilitated a decrease in curie temperature from 400 to 193°C and an increase in peak dielectric constant. Similar behaviour has been observed in different perovskite

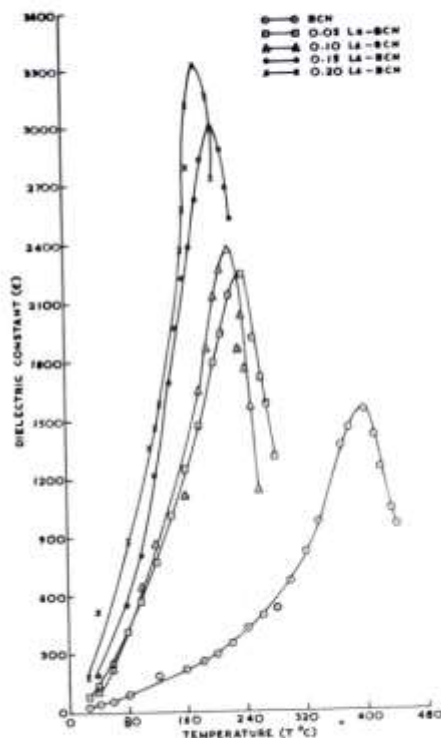


Fig. 4 - The variation of dielectric constant with temperature for modified and unmodified BCN

systems like barium strontium titanate and barium calcium titanate⁸. Curie-Weiss law has been obeyed in the para region of the compositions, 6 to 10, and the curie constants have been computed for all the compositions. Curie constant of the order of 10^5 K has been observed in all the five materials.

In both the systems BCT and BCN, the decrease in curie temperature with increase of lanthanum may be attributed to the decrease in polarizability of A-site cations (for example $\alpha(\text{Ba}^{2+}) = 1.55 \times 10^{-24} \text{ cm}^3$, while $\alpha(\text{La}^{3+}) = 1.04 \times 10^{-24} \text{ cm}^3$). Under these circumstances the dipoles at the large cation site interact weakly with those at the Ta^{5+} and Nb^{5+} sites in BCT and BCN, respectively, resulting in a decrease of T_c . Similar phenomenon has been reported for the systems $(\text{Na}_{1-x}\text{La}_x)_2\text{Pb}_{1-x}\text{TiO}_3$ and $(\text{Li}_{1-x}\text{La}_x)_2\text{Pb}_{1-x}\text{TiO}_3$ [Refs 9, 10].



Table 2 - Dielectric, DTA and resistivity data of lanthanum doped BCT and BCN

Composition	Curie temp.		ϵ_{RT}	ϵ_{TC}	$\delta, ^\circ\text{C}$	$\rho_{RT} \times 10^8$ $\Omega\text{-cm}$	$\rho_T \times 10^8$ $\Omega\text{-cm}$	$T, ^\circ\text{C}$	$E_g (\text{eV})$	
	$T_c, ^\circ\text{C}$	$T_D, ^\circ\text{C}$							+ve	-ve
BCT	319	300.1	42	779	70.0	4.80	11.90	75	0.11	0.11
0.05 La-BCT	270	297.3	28	203	73.0	2.30	7.13	60	0.17	0.12
0.10 La-BCT	244	—	23	462	73.5	0.16	3.97	55	0.21	0.12
0.15 La-BCT	221	—	23	493	52.0	0.47	1.76	55	0.27	0.26
0.20 La-BCT	199	—	26	423	40.0	0.25	0.88	55	0.15	0.14
$C \times 10^3 \text{ K}$										
BCN	400	426.3	42	1531	0.8	0.36	5.02	68	0.12	0.11
0.05 La-BCN	242	289.8	48	2279	1.5	0.19	0.38	60	0.21	0.21
0.10 La-BCN	226	—	52	2357	1.2	0.21	0.29	50	0.04	0.05
0.15 La-BCN	216	—	57	3167	1.8	0.18	0.23	50	0.02	0.06
0.20 La-BCN	193	—	56	3317	32.0	0.14	0.18	50	0.08	0.07

The room temperature and peak temperature dielectric constants, ϵ_{RT} , ϵ_{TC} respectively, curie temperatures obtained from both dielectric measurements (T_c) and DTA (T_D), the intensity of diffusivity (δ) and curie constants (C) for pure and La-modified BCT and BCN samples are given in Table 2.

4.2 DC Resistivity Properties

Variation of dc resistivity with temperature in unmodified and modified BCT and BCN has been studied in the temperature range from room temperature to 230°C by applying a small voltage of 10 V. The dc resistivity versus temperature for the two systems is depicted in Figs 5 and 6.

It is obvious from these figures that an anomalous behaviour is evident in both the systems. That is, there is an initial increase of resistivity to a maximum at a particular temperature, followed by a gradual decrease. This type of positive temperature coefficient (PTC) anomalous behaviour in ferroelectrics can be explained by Heywang¹¹ model when the temperature corresponding to the maximum resistivity coincides with their respective T_c values. But, in the present study the temperatures corresponding to the maximum resistivity are well below their respective T_c values in unmodified and modified BCT as well as in BCN. One of the reasons for this behaviour may be due to the vacancy of barium sites in BCT and BCN as a result of aliovalent lanthanum doping in place of Ba^{2+} . However, this type of behaviour

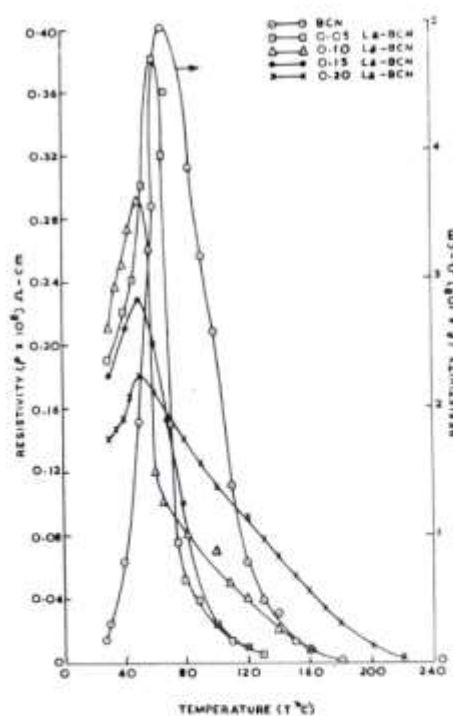


Fig. 5 - DC resistivity versus temperature for modified and unmodified BCT

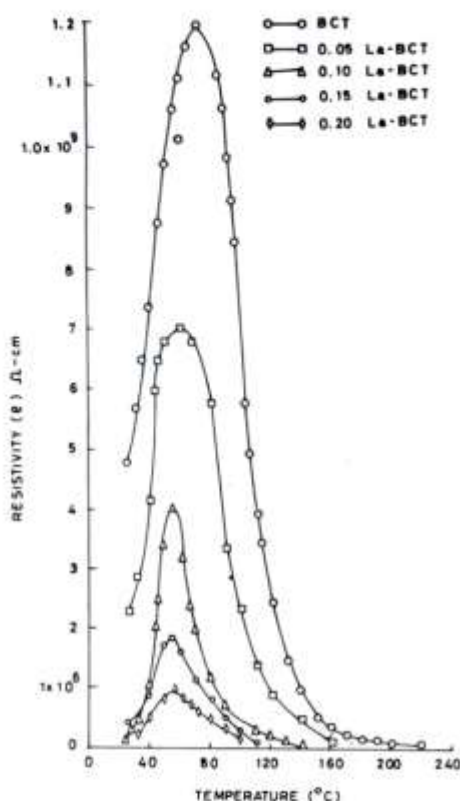


Fig. 6 - DC resistivity versus temperature for modified and unmodified BCN

has also been observed in other systems such as Dy-doped barium sodium niobate ceramics¹². The room temperature resistivity (ρ_{RT}) and peak resistivity (ρ_T) are found to decrease with the increase in lanthanum concentration in BCT. All these curves show resistivity maximum at a certain range of temperature, which is a property of a semiconducting material and doping of lanthanum either in BCT or in BCN did not alter this behaviour.

The variation of resistivity (ρ) with temperature (T) in BCT and BCN obeys the equation.

$$\rho = \rho_0 \exp(E_a/kT)$$

where E_a is the activation energy, k is the Boltzmann constant and ρ_0 is a constant. The values of E_a in both the regions of all the compositions have been computed from the above equation and tabulated in Table 2, along with ρ_{RT} , ρ_T and temperature of maximum resistivity (T_m , °C).

5 Conclusions

In view of the above observations, it is concluded that, the systems BCT and BCN are monophasic and tetragonal. Substitution of lanthanum in BCT as well as in BCN to an extent of $x=0.100$ in $\text{Ba}_{1-x}\text{La}_x(\text{Cu}_{1/3}\text{Ta}_{2/3})\text{O}_3$ and $\text{Ba}_{1-x}\text{La}_x(\text{Cu}_{1/3}\text{Nb}_{2/3})\text{O}_3$ did not alter the crystal structure. Nevertheless progressive addition of lanthanum caused a decrease in T_c with a simultaneous increase in peak dielectric constant. BCT and La-modified BCT samples exhibited diffused phase transition. Both the compositions of BCT and BCN showed resistivity maxima with a PTCR behaviour. Addition of lanthanum did not show any significant effect on the conductivity properties of BCT and BCN.

Acknowledgement

One of the authors (KSR) is grateful to CSIR, New Delhi, for their financial support.

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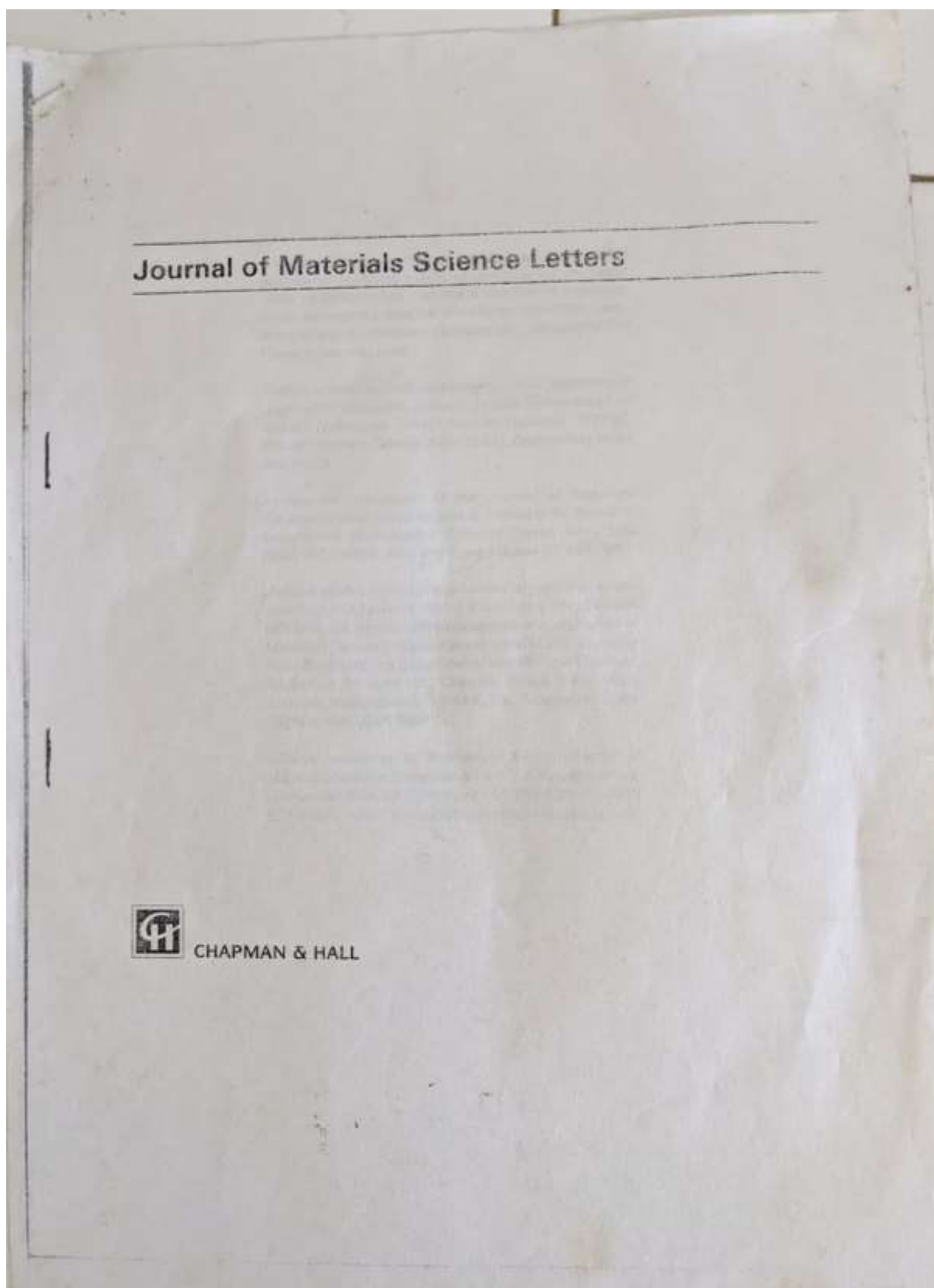
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Dielectric and conductivity properties of lanthanum-modified strontium copper niobate

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A number of compounds with perovskite structure exhibit ferroelectricity. Barium titanate, one of the important ferroelectric materials for device applications [1], has a relatively low Curie temperature (T_c) of 120 °C. In order to alter the Curie point to higher or lower temperatures, solid solutions are usually formed by judicious selection of suitable substituents either in Ba or in Ti sites. For instance, modest amounts of Pb^{2+} in $BaTiO_3$ raise its T_c whereas Sr^{2+} for Ba or Zr for Ti lowers the T_c [2-4]. Investigations on the synthesis of new perovskite-type compounds led to the unravelling of many compositions [5-7], some of which, particularly copper-containing Perovskites [6], were found to exhibit higher Curie temperatures in the range 300-1200 °C. Since the ferroelectric phase transition is connected with the electron-phonon interaction which takes place in the presence of the Jahn-Teller pseudo-effect, this is very pronounced in compounds containing bivalent copper of $3d^9$ electron configuration. Strontium copper niobate [$Sr(Cu_{1-x}Nb_x)O_3$ (SCN)] crystallizes in a tetragonal symmetry with lattice parameters $a = 0.3944$ nm and $c = 0.4074$ nm, and is also reported to be a ferroelectric material with Curie temperature of 390 °C [6]. No further studies on SCN have appeared in the literature. In this letter we report the synthesis, characterization and the effect of partial substitution of lanthanum on the dielectric and conductivity properties of SCN.

Samples of SCN and lanthanum-incorporated strontium copper niobate (La-SCN) were prepared by the usual ceramic method with $SrCO_3$, CuO , Nb_2O_5 and La_2O_3 of AR grade purity as starting materials. Stoichiometric quantities of these materials are taken to yield the compositions $(Sr_{1-x}La_x)(Cu_{1-x}Nb_x)O_3$ with $x = 0.0, 0.075, 0.150, 0.225$ and 0.300 . The physical mixtures were dry ground into fine powders and these were further mixed thoroughly in presence of alcohol to improve the homogeneity. The mixtures were calcined twice at 950 °C for 4 h. The calcined samples were pressed into pellets with 5% poly(vinyl acid) as binder using a pressure of 4.5 tonne cm^{-2} (68 MPa). The pelletized samples were sintered at 1200 °C for 6 h. The phase identification was followed by X-ray diffraction (XRD) using $CuK\alpha$ radiation. The dielectric

measurements were done at 1 kHz using a capacitance bridge, digital LCR meter (type VLCR6). D.c. resistivity measurements on these samples were performed by the two-probe method with a ECIL electrometer amplifier (type EA815) using a d.c. voltage of 10 V. Silver-coated pellets were used for the above studies.

From the XRD data, SCN and La-SCN samples were found to be monophasic and tetragonal in symmetry in accordance with earlier reports [6, 7]. The lattice parameters for these samples are given in Table I. As can be seen from the XRD data in Table I, substitution of La for Sr seems to promote the tetragonal distortion.

Fig. 1 shows the temperature (T) dependence of the dielectric constant (ϵ) of SCN and La-SCN samples. As can be seen, increased substitution of Sr^{2+} by La^{3+} resulted in a reduction of the Curie temperature, from 345 to 245 °C, with an increase in peak dielectric constant from 255 to 1439. Furthermore, higher polarization (α) of La might be responsible for the enhanced dielectric constant [$\alpha(Sr^{2+}) = 0.86$ and $\alpha(La^{3+}) = 1.08$]. Similar behaviour was reported for other systems [8] such as $Ba(Ti_{1-x}Zr_x)O_3$. Since the maximum value of ϵ at the Curie point T_c in an ideal ferroelectric crystal can be described by the Curie-Weiss law, $\epsilon = C/(T - T_c)$ for $T > T_c$. The Curie constant (C) can be obtained from the temperature dependence of ϵ at $T > T_c$. The estimated Curie constants for these samples are given in Table II, together with the respective Curie temperatures, room-temperature dielectric constant (ϵ_{RT}) and peak dielectric constant (ϵ_{TC}).

Fig. 2 shows the variation of resistivity (ρ) with temperature for SCN and La-SCN. These samples show resistivity maxima at temperatures (ρ_T) which

TABLE I Lattice parameters of the SCN and La-SCN samples

Sample	<i>a</i> (nm)	<i>c</i> (nm)	<i>c/a</i>
$Sr(Cu_{1-x}Nb_x)O_3$	0.392 63	0.411 81	1.046
$(Sr_{0.925}La_{0.075})(Cu_{1-x}Nb_x)O_3$	0.392 85	0.412 28	1.049
$(Sr_{0.85}La_{0.15})(Cu_{1-x}Nb_x)O_3$	0.383 39	0.420 58	1.091
$(Sr_{0.775}La_{0.225})(Cu_{1-x}Nb_x)O_3$	0.386 29	0.423 15	1.095
$(Sr_{0.7}La_{0.3})(Cu_{1-x}Nb_x)O_3$	0.381 06	0.420 25	1.102